



Shape-Driven $\text{Fe}_x\text{O}_{x+1}$ ($x = 2, 3$) Nanostructures for Magnetic Hyperthermia Applications

Haripal Singh Dhayal¹, Prabhavati C. Sutar², Rohit R. Koli³, P. V. Raghavendra⁴, S. D. Kaushik⁵,
Dhiraj Bhatia⁶, and Nishad G. Deshpande^{1,*}

¹Department of Physical and Applied Sciences, Indian Institute of Information Technology,
Surat-394190 (Gujarat) India

²Thick and Thin Film Device Laboratory, Department of Physics, Shivaji University, Kolhapur-
416004 (M.S.) India

³Department of Applied Chemistry, National Yang Ming Chiao Tung University, Hsinchu
300093, Taiwan.

⁴Department of Education in Science & Mathematics, National Council of Educational Research
and Training, New Delhi, India

⁵UGC-DAE Consortium for Scientific Research, BARC Campus, Trombay Mumbai – 400085
(M.S.) India

⁶Department of Biological Sciences and Engineering, Indian Institute of Technology
Gandhinagar (IITGN), Gandhinagar, Gujarat, India

*Corresponding author: nishad.deshpande@iiitsurat.ac.in

Abstract

Shape-driven is a key factor that affects nanostructure materials and its response to magnetic fields and its energy loss. This makes it a useful design parameter for biomedical and nano-actuation applications. In this study, we present the methodical synthesis and thorough examination of shape-engineered $\text{Fe}_x\text{O}_{x+1}$ ($x = 2, 3$) nanodisks (NDs) for regulated magnetic hyperthermia applications. By carefully controlling the ND shapes and oxidation states of iron oxide phases, the specific magnetic and hyperthermia properties was controlled. The engineered NDs show better heating efficiency when exposed to alternating magnetic fields. This is because of the combined effects of Néel and Brownian relaxation processes, which are controlled by shape-induced anisotropy. In-depth structural, morphological, and magnetic analyses show that shape, magnetic loss mechanisms, and functional performance are all closely linked. These $\text{Fe}_x\text{O}_{x+1}$ NDs have many uses, making them promising, earth-abundant, and biocompatible platforms for the next generation of magnetic hyperthermia therapies and remotely driven nano-actuation systems.

Keywords: Iron oxide, magnetic materials, shape anisotropy, hyperthermia, magnetic actuation, and biomedical applications.

1. Introduction

The rising prevalence of cancer has necessitated the advancement of innovative therapeutic modalities beyond traditional chemotherapy and radiotherapy. Magnetic hyperthermia has emerged as a promising non-invasive treatment, wherein magnetic nanoparticles (MNPs) generate localized heat under an alternating magnetic field (AMF) to selectively destroy tumor cells [1-4]. The heating efficiency in magnetic hyperthermia is determined by various interrelated factors, such as particle size, shape anisotropy, saturation magnetization, and field conditions [4-6]. Many physical and chemical methods have been tried to make effective MNPs, but using shape-controlled agents has gotten a lot of attention because it can control and keep shape properties, give good hyperthermia results, is good for the environment, and is more biocompatible [5-9]. Fe₃O₄ nanoparticles are especially appealing for biomedical uses because they have a high saturation magnetization, can generate a lot of heat, are not very toxic, and are clinically relevant. Previous research demonstrates that Fe₃O₄ nanoparticles approximately 10–22 nm achieves the most favorable specific absorption rate (SAR) [10-12], while particles smaller than 10 nm or exceeding ~35 nm exhibit diminished heating efficiency [10,11,13]. In addition to size effects, shape engineering of Fe₃O₄ nanostructures is very important for improving hyperthermia performance by increasing magnetic anisotropy, relaxation losses, and hysteresis area [4,14]. The phase change from weakly magnetic Fe₂O₃ to ferrimagnetic Fe₃O₄ is also very important. This is because Fe₃O₄ has a higher saturation magnetization, mixed-valence Fe²⁺/Fe³⁺ states, and better magnetic relaxation mechanisms that are needed to make heat work. Its lower Curie temperature also helps with better thermal regulation during treatment, which lowers the risk of overheating healthy tissues nearby [15–18]. Fe₃O₄ is also clinically accepted and FDA-approved for biomedical use because it breaks down naturally, is not very toxic, and is easy to functionalize on the surface [19–21].

Taking into account the above factors, this study focuses on shape-engineered Fe₂O₃ nanodisks (NDs) that were then changed into Fe₃O₄ through a controlled annealing process that didn't change their shape much. This strategy allows for the preservation of shape-induced properties while enhancing magnetic properties, leading to improved magnetic hyperthermia performance, which is elaborated upon in detail.

2. Synthesis Methods

2.1 Synthesis Procedure

Shape-engineered Fe_xO_{x+1} ($x = 2, 3$) NDs were synthesized through a solvothermal method. In a standard protocol, Fe(NO₃)₃ was solubilized in 80 mL of ethanol, subsequently followed by the incorporation of 1 mL of oleylamine. The solution was then put into a Teflon-lined autoclave and heated to 200 °C for solvothermal treatment. A red Fe₂O₃ NDs, which was then washed several times with ethanol by centrifugation and dried in a vacuum. After being made, the Fe₂O₃ NDs were changed into Fe₃O₄ NDs by annealing in nitrogen ambient at 500 °C for 3 hours. All the chemicals

were purchased through Sigma Aldrich with AR grade and no further purification is done. Different physico-chemical characterization methods were used to confirm the formation of the black Fe_3O_4 phase.

2.2 Characterization Methods

An X-ray diffractometer (Bruker AXS D8) was used to do structural characterization. Field-emission scanning electron microscopy (FESEM) (TESCAN, Mira3) was used to look at the samples' morphology, and energy-dispersive X-ray spectroscopy (EDS) was used to look at their elemental composition. The crystallinity as well as crystalline phase was done using TEM and SAED studies (FEI Tecnai G2-F20). X-ray photoelectron spectroscopy (XPS) (Thermo Fisher Scientific ESCALAB Xi+) was used to learn more about the oxidation states and the composition of the surface elements. A neutron diffraction study was done to back up the structural studies. We used a magnetic hyperthermia setup (Easy Heat 8310, Ambrell) to do magnetic hyperthermia studies.

3. Results and Discussion

Transmission electron microscopy (TEM) and selected area electron diffraction (SAED) was used to study the as-synthesized Fe_2O_3 and the thermally transformed Fe_3O_4 , which is as shown in Fig. 1[(a)-(d)]. The results of TEM and SAED revealed that the $\alpha\text{-Fe}_2\text{O}_3$ nanostructures that were grown are polycrystalline in nature [Fig. 1 (a)-(b)].

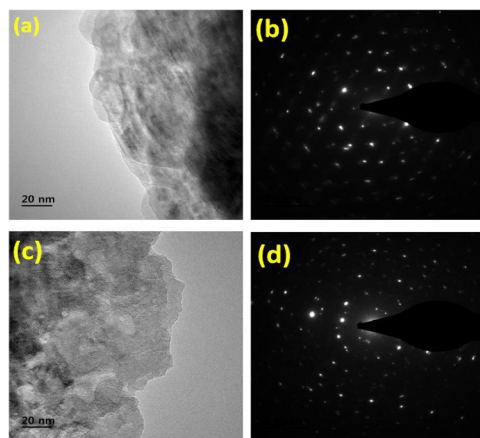


Figure 1. TEM micrographs of (a) Fe_2O_3 NDs and (c) Fe_3O_4 NDs, as well as the corresponding SAED patterns of (b) Fe_2O_3 and (d) Fe_3O_4 .

When these α -Fe₂O₃ nanostructures are annealed under inert atmosphere, they get transformed into Fe₃O₄, as shown by the changes in the diffraction patterns as is evident from Fig. 1[(c)-(d)]. The high-resolution images obtained under HRTEM mode reveals aggregation of small nanocrystals in Fe₂O₃ NDs structure [Fig. 1(a)] while after annealing in inert atmosphere, the Fe₃O₄ phase NDs structure become more compact by recrystallization due to energy provided due to high temperature heating [Fig. 1(c)].

Further, Field-emission scanning electron microscopy (FESEM) images [Fig. 2(a)-(b)] was analyzed that demonstrated the emergence of distinctly defined nanodisk-like morphologies having diameter \sim 220 nm [Fig. 2(a)] of Fe₃O₄ NDs. High-magnification FESEM images [Fig. 2(b)] showed that these NDs are made up of smaller nanoparticles that come together and arrange themselves to make the disk-like structure. Hierarchical NDs structures have a lot of surface-area and good magnetic and structural properties, which makes them very useful for biomedical uses like magnetic hyperthermia, targeted drug delivery, and bioimaging.

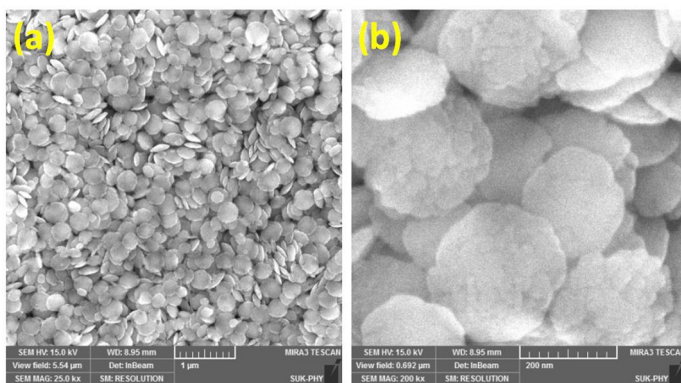


Figure 2. FESEM micrographs of Fe₃O₄ NDs: (a) low-magnification image and (b) its magnified view.

The elemental analysis through EDS studies [Fig. 3(a)] of Fe₃O₄ nanostructures verifies the presence of Fe and O elements, devoid of detectable impurities, and exhibiting concentrations that matches with the anticipated stoichiometry of Fe₃O₄. Moreover, the XRD pattern in Figure 3(b) confirms formation of pure Fe₃O₄ structure with cubic, Fd-3m space group (JCPDS card 00-001-1111). The major reflection for planes (220), (311), (400), (511) and (440) were observed at 2θ values 30.08°, 35.52°, 43.34°, 57.11° and 62.78°, respectively.

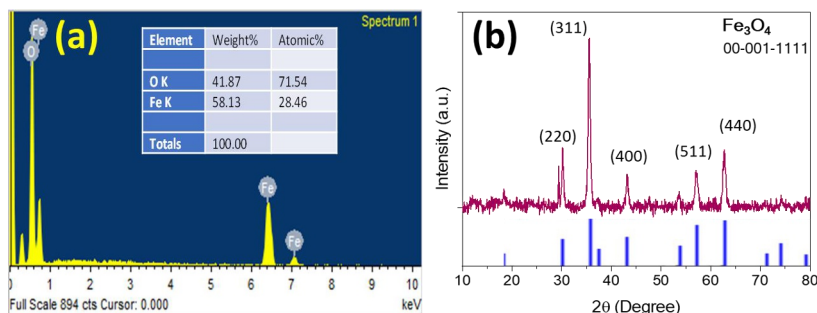


Figure 3. (a) Energy-dispersive X-ray spectroscopy (EDS) spectra of Fe_3O_4 NDs samples; (b) XRD pattern of Fe_3O_4 NDs.

To further validate the elemental studies, we used X-ray photoelectron spectroscopy (XPS) on the Fe_3O_4 samples. Figure 4(a) shows the survey XPS spectrum of the Fe_3O_4 nanostructures. It shows that the Fe $2p$, O $1s$, and C $1s$ signals are all present at their normal binding energy levels. Figure 4(b) shows the high-resolution Fe $2p$ spectra of the sample. It has two main peaks at 712 eV and 725 eV, which are Fe $2p_{3/2}$ and Fe $2p_{1/2}$, respectively. Whereas, after deconvolution the peaks at 709.97 eV and 723.85 eV are ascribed to Fe^{3+} state while the peaks observed at 711.52 eV and 726.16 eV are related to Fe^{2+} state. Figure 4(c) displays the high-resolution O $1s$ XPS spectra of the samples. Two main parts can be seen in the binding energy ranges of 529–531 eV and 532–535 eV. Deconvolution shows that there are contributions from lattice oxygen (O_{lat}) and surface-adsorbed oxygen species (O_{ads}), which means that there is more than one type of oxygenated species on the Fe_3O_4 surface.

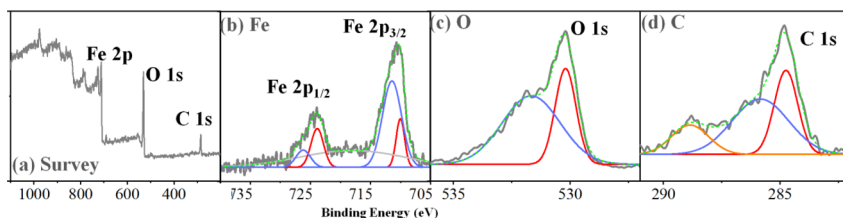


Figure 4. X-ray photoelectron spectroscopy (XPS) analysis of Fe_3O_4 nanostructured samples: (a) survey spectrum and high-resolution spectra of (b) Fe $2p$, (c) O $1s$, and (d) C $1s$.

The peak at about 529.97 eV is due to lattice oxygen (O_2), and the higher binding energy components (531.36–533.81 eV) are due to surface-adsorbed oxygen species like hydroxyl and carbonate-related species, which is in line with what has been reported in the literature [22, 23]. The XPS spectra of C $1s$, depicted in Fig. 4(d), was deconvoluted into three peaks at 284.8, 285.9,

and 288.9 eV, corresponding to C-C, C-OH, and C=O, respectively. The C-C peak is mostly ascribed to the carbonaceous elements from the CB, whereas C-O and C=O are attributed to the partly dehydrated residues on the nanocomposite's surface.

The prepared Fe₃O₄ NDs is then utilized for magnetic hyperthermia in high frequency AC magnetic field. Figure 5 shows the results heat generation profile of different amounts of these NDs in 1mL DI water taken in vials. For effective in vivo magnetic hyperthermia, it is essential that a minimal quantity of dispersed Fe₃O₄ nanoparticles can elevate the temperature to the therapeutic range (~43 °C). The heating performance of the synthesized Fe₃O₄ nanostructures was assessed at varying concentrations of 1, 3, and 5 mg mL⁻¹ under alternating magnetic field (AMF) strengths of 335.2 Oe. Figure 5 shows how the temperature changed during the first 20 minutes of induction heating. Temperature–time kinetic plots showed that 3 mg mL⁻¹ has a stable and controlled temperature rise between 40°C and 43 °C, which is the best range for treating hyperthermia. On the other hand, temperature profiles for other concentration samples are not regular and are less consistent. The SAR values of prepared magnetic NDs were calculated using following equation-

$$\text{SAR} = c \cdot \left(\frac{m_s}{m_m} \right) \cdot \left(\frac{\Delta T}{\Delta t} \right)_{\text{initial}} \quad (1)$$

where c represents the specific heat capacity of the sample suspension (for the specified suspension, $c = 4.186 \text{ J}/(90 \text{ }^\circ\text{C})$), (m_m/m_s) denotes the mass fraction of the sample in the suspension, and $(\Delta T/\Delta t)_{\text{initial}}$ indicates the beginning slope of the time-dependent temperature kinetic curve.

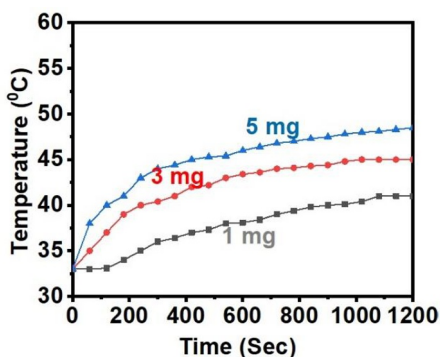


Figure 5. Temperature kinetic graphs for water-dispersed Fe₃O₄ samples with different amounts of Fe₃O₄ nanostructures and different strengths of AMF at a frequency of 276 kHz.

Table 1 gives comparative study of performance of prepared NDs sample with previously reported ferrite samples having different shapes. The optimum heat generation capacity comparison was

done by studying specific absorption rates (SAR) of these materials. The optimized Fe₃O₄ NDs with 3mg amount showed high SAR value of 1043.7 W/g which is significantly higher than most of previously studies ferrites as depicted in table 1.

Material	Shape	Size (nm)	Frequency (kHz)	SAR (W/g)	Ref.
Fe ₃ O ₄	Cube	26	70	170	24
FeO@Fe ₃ O ₄	Core-Shell	23	200	310	25
Fe ₃ O ₄	Nanoring	125	488	5000	26
Mn _{0.5} Co _{0.5} Fe ₂ O ₄	Hollow-Spherical	220	300	77.7	27
Fe ₃ O ₄	Nanoflower	200	300	79.1	28
Fe₃O₄	Nanodisks	220	280	1043.7	This Work

Table 1 Comparison of SAR values of different magnetic materials.

Conclusions

Shape-driven magnetic iron oxide nanodisks is obtained successfully through a solvothermal process followed by annealing which changed polycrystalline α -Fe₂O₃ to Fe₃O₄. Structural and morphological analyses validated the emergence of uniformly distributed ND structures formed from aggregated nanoparticles, providing a high surface-area and advantageous magnetic characteristics. Studies on magnetic hyperthermia showed that heating behavior depends on concentration. In particular, ~3 mg mL⁻¹ Fe₃O₄ NDs showed a stable temperature rise in the therapeutic range (40–43 °C) at low magnetic fields. These results showed importance of precise control over phase, shape, and morphology when making iron oxide NDs that work well for biomedical applications that use hyperthermia.

Acknowledgement

Authors are thankful to the UGC-DAE CSR, Mumbai for proving financial support through Collaborative project scheme (*Ref: CRS/2021-22/03/590 dated 17-09-2025*). Author HSD is thankful to the Indian Institute of Information Technology Surat, Gujarat, India, for providing the institute's research Fellowship. NGD is thankful to Indian National Science Academy (INSA), New Delhi for providing INSA visiting scientist award (*INSA/SP/VSP-68/2025-26/314 dated 22 September 2025*). Additionally, authors are thankful to the ANRF-PAIR program (ANRF/PAIR/2025/000008/PAIR-B dated 12 July, 2025) for financial support.

References

- [1] Uri D, Kruse A M, Meenach S A, Acta Biomater. 10 (2014) 2622.

- [2] Milla M, Yu S M, Laromaine A, Chem. Eng. J. 340 (2014) 173.
- [3] Rodovalho F L, Capistrano G, Gomes J A, Sodré F F, Chaker J A, Campos A F C, Bakuzis A F, Sousaa M H, Chem. Eng. J. 302 (2016) 725.
- [4] Kolhatkar A G, Jamison A C, Litvinov D, Willson R C, Lee T R, Int. J. Mol. Sci. 14 (2013) 15977.
- [5] Weng X, Guo M, Luo F, Chen Z, Chem. Eng. J. 308 (2017) 904.
- [6] Ahmad N, Sharma S, Alam M K, Singh VN, Shamsi S F, Mehta B R, Fatma A, Colloids Surfaces B: Biointerfaces 81 (2010) 81.
- [7] Haverkamp R G, Marshall A T, J. Nanoparticle Res. 11 (2009) 1453.
- [8] Shankar S S, Rai A, Ahmad A, Sastry M, J. Colloid Interface Sci. 275 (2004) 496.
- [9] Gong J L, Wang X Y, Zeng G M, Chen L, Deng J H, Zhang X R, Niu Q Y, Chem. Eng. J. 185 (2012) 100.
- [10] Jeun M, Lee S, Kang J K, Tomitaka A, Kang K W, Kim Y I, Takemura Y, Chung KW, Kwak J, Bae S, Appl. Phys. Lett. 100 (2012) 3.
- [11] Fortin J P, Wilhelm C, Servais J, Ménager C, Bacri J C, Gazeau F, J. Am. Chem. Soc. 129 (2007) 2628.
- [12] Lartigue L, Innocenti C, Kalaivani T, Awwad A, Duque M D M S, Guari Y, Larionova J, Guefrin C, Montero J L G, Montero V B, Arosio P, Lascialfari A Gatteschi D, Sangregorio C, J. Am. Chem. Soc. 133 (2011) 10459.
- [13] Mornet S, Vasseur S, Grasset F, Duguet E, J. Mater. Chem., 14 (2004) 2161.
- [14] Chung S H, Hoffmann A, Bader S D, Liu C, Kay B, Makowski L, Chen L, Appl. Phys. Lett. 85 (2004) 2971.
- [15] Makarov V V, Makarova S S, Love A J, Sinitsyna O V, Dudnik A O, Yaminsky I V, Taliansky M E, Kalinina N O, Langmuir 30 (2014) 5982.
- [16] Boubeta C M, Simeonidis K, Serantes D, Leborán I C, Kazakis I, Stefanou G, Peña L, Galceran R, Balcells L, Monty C, Baldomir D, Mitrakas M, Angelakeris M, Adv. Funct. Mater. 22 (2012) 3737.
- [17] Roca A G, Lopez-Barbera J F, Lafuente A, Özel F, Fantechi E, Muro-Cruces J, Hémadi M, Sepulveda B, Nogues J, Phys. Reports 1043 (2023) 1.
- [18] Bhakar S K, Sharma B, Satapathy S, Deshmukh P, Srihari V, Singh R, Majumder S K, BioNanoScience 15 (2025) 508.
- [19] Bobo D, Robinson K J, Islam J, Thurecht K J, Corrie S R, Pharm. Res 33 (2016) 2373.

- [20] Revia R A and Zhang M, *Mater. Today* 19 (2016) 157.
- [21] Gossuin Y, Gillis P, Hocq A, Vuong Q L, Roch A, *Wiley Interdiscip. Rev. Nanomed. Nanobiotechnol* 1 (2009) 299.
- [22] Wilson D, Langell M A, *Appl. Surf. Sci.* 303 (2014) 6.
- [23] Amin M O, D'Cruz B, Madkour M, Al-Hetlani A, *Microchimica Acta* 186 (2019) 503.
- [24] Marciello M, Connord V, Verdaguer S V, Vergés M A, Carrey J, Respaud M, Serna C J, Morales M P, *J. Mater. Chem. B* 1 (2013) 5995.
- [25] Lak A, Cassani M, Mai B T, Winckelmans N, Cabrera D, Sadrollahi E, Marras S, Remmer H, Fiorito S, Jimeno L C, Litterst F J, Ludwig F, Manna L, Teran F J, Bals S, Pellegrino T, *Nano Lett.* 18 (2018) 6856.
- [26] Yang Y, Liu X, Lv Y, Herng T S, xu X, Xia W, Zhang T, Fang J, Xiao Wen, Ding J, *Adv. Funct. Mater.* 25 (2015) 812.
- [27] Shen K, Li L, Tan F, Wu S, Jin T, You J, Chee M Y, Yan Y, Lew W S, *Nanoscale* 15 (2023) 17946.
- [28] Fan Y, Li Y, You J, Shen K, Chen W, Li L, *Materials Chemistry and Physics* 329 (2025) 130045.

Open Access This chapter is licensed under the terms of the Creative Commons Attribution-NonCommercial 4.0 International License (<http://creativecommons.org/licenses/by-nc/4.0/>), which permits any noncommercial use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license and indicate if changes were made.

The images or other third party material in this chapter are included in the chapter's Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the chapter's Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder.

